

# Molecular Dynamics probing of the energy spectrum of particles in radiation stimulated processes

A. Kiv, Department of Materials Engineering, Ben-Gurion University of the Negev, Israel; N. Mykytenko, Hewlett-Packard, Yehud, Israel; D. Fuks, Department of Materials Engineering, Ben-Gurion University of the Negev, Israel; I. Dahan, Nuclear Research Center-Negev, POB 9001, Israel; L. Meshi, Department of Materials Engineering, Ben-Gurion University of the Negev, Israel

## Abstract

The approach based on the classical molecular dynamics (MD) is developed that allows to probe the energy spectrum of particles in radiation induced processes. To simulate the effect of particles collisions in the selected interval of the energy spectrum the “shock function” is introduced to the standard scheme of MD. This function describes the forces acting on the lattice atoms by the incident particles in the selected energy interval. The approach is illustrated by modeling the ion bombardment of triatomic model crystal with significantly different atomic masses of constituents. It can be useful in particular for a prediction of clusters type defects formed in polyatomic crystals.

**Keywords:** Molecular Dynamics; computer modeling; cluster defects in solids.

## 1. Introduction

Molecular dynamics (MD) is a widely used method in the investigations of radiation induced processes. This method is especially important when the experimental studies are time-consuming and complex, for example in the case of exposure to reactor radiation. MD calculations allow tracking the trajectories of particles, studying mechanisms of materials destruction and finding the equilibrium configurations of structural defects [1-8]. Due to the application of MD method the peculiarities of the passage of fast particles through the matter have been established.

We developed the MD software that allows studying radiation induced processes in the selected range of energy spectrum of incident particles. Mechanisms of formation of radiation defects essentially depend on the type of incident particles and their energy. As a rule at the long-term unchanging irradiation conditions a stationary energy distribution of incident particles is set. In each interval of energy spectrum of these particles the specific mechanisms of defect formation are realized. The final radiation effect is determined by a superposition of radiation effects caused by incident particles in all intervals of their energy spectrum. Therefore to decrypt the mechanisms of formation of radiation defects it is important to clarify the mechanisms of

radiation-stimulated processes in different intervals of the energy spectrum of incident particles.

To simulate the effect of atomic collisions in the selected interval of the energy spectrum the “shock function” is introduced. This function determines the forces ( $F_{SH}$ ) acting on the lattice atoms as described earlier in [9]. The pulses that are transferred to lattice atoms during irradiation are characterized by special random function. This function shows which atom in the irradiated sample is knocked, what energy value is chosen from the selected energy interval, and what the direction of hit is.

It should be noted that due to the superposition of radiation effects caused by incident particles from different intervals of the energy spectrum, the result of the action of particles from selected energy range can be partly or fully hidden. Therefore to estimate the role of different mechanisms in the radiation induced destruction of material it is necessary using the proposed approach to carry out the suggested simulations in different intervals of the energy spectrum and take into account the contribution of each energy interval to the final radiation effect.

The developed approach is especially useful when the compound under irradiation consists of atoms with significantly different masses. For example we demonstrate that it is possible to obtain new information regarding the formation of defects in compounds that are used as nuclear materials ( $U(Al, Si)_3$ ). Another example is a possibility to form the appropriate track structures for creation of electronic devices in track electronics [9 – 13].

The properties of track devices depend on the shape of tracks and the electronic structure of the internal track surfaces. The necessary properties of tracks can be obtained using the proposed approach.

## 2. Description of the approach

If radiation effects induced by the incident particles in the energy range ( $E_1, E_2$ ) of the total energy spectrum are studied, the maximal energy transferred in elastic collisions to the lattice atom of mass  $M$  by ions with mass  $m_{ion}$  is expressed as:

$$\varepsilon_{1,2} = \frac{4m_{ion}M}{(m_{ion} + M)^2} \cdot E_{1,2}, \quad (1)$$

In (1) the transferred energies  $\varepsilon_1$  and  $\varepsilon_2$  correspond to the energies of incident particles  $E_1$  and  $E_2$ .

To simulate the elementary process induced by incident particles a random function (RF) is used. RF inserted to the computer program performs three tasks:

- Selects an atom in the target lattice which gets a hit;
- Selects an energy value from the interval  $(\varepsilon_1, \varepsilon_2)$ ;
- Selects an orientation of the pulse transferred to the target atom.

A linear congruent generator (LCG) for RF is an algorithm that yields a sequence of pseudo-randomized numbers calculated with a discontinuous piecewise linear equation. This generator presents one of the best-known pseudorandom number generator algorithms that are easily implemented and fast, especially on computer hardware which can provide modulo arithmetic by storage-bit truncation. These generators based on linear congruent method are especially useful for non-cryptographic applications, such as modeling. They are effective and most used in empirical tests and show good statistical characteristics. The generator described in [14 – 16] is defined by the recurrence relation:

$$X_{n+1} = (aX_n + c) \pmod{m} \quad (2)$$

where  $X_0$  is an initial value. In our model the parameters are:  $m = 232$ ,  $a = 214013$ ,  $c = 2531011$ .

It is assumed in the suggested model that the number of atoms simultaneously subjected to shock is proportional to the dose rate, and the number of steps is proportional to the irradiation dose. In order to determine the kinetic energy transferred to the lattice atom the scaling of the forces  $F_{SH}$  should be implemented. As a reference point we used the energy that is necessary for irreversible displacement of lattice atom to interstitial position in elastic collision ( $E_d$ ). Then the following relation may be written:

$$\frac{(F_{SH}t)^2}{2M} = E_d, \quad (3)$$

where  $t$  is duration of the action of the force in the process of one hit.

In computer experiment we gradually increased the value of the force  $F_{SH}$  to the point where the atoms begin to move irreversibly to interstitial positions. This magnitude of the force  $F_{SH}$  corresponds to  $E_d \approx 25 - 30$  eV. Such procedure allows to define the interval for  $F_{SH}$  that corresponds to the selected interval  $(\varepsilon_1, \varepsilon_2)$  and respectively  $(E_1, E_2)$  as determined by Eq. (1).

### 3. Computer simulation of ion bombardment of polyatomic crystals

In the MD method [17-19] the classical equations of motion with an appropriate potential of the interaction between particles are solved. Verlet algorithm [20] is usually used to solve these equations. To simulate the action of the shock function, the total force acting on atom  $i$  is presented by expression:

$$\vec{F}_i^n = -\sum \frac{\partial \Phi(\vec{r}_{ij})}{\partial x_i} + \vec{F}_{SH}, \quad (4)$$

where  $\Phi(\vec{r}_{ij})$  is the interatomic potential and  $r_{ij} = |\vec{r}_i - \vec{r}_j|$  is the distance between atoms  $i$  and  $j$ .

To apply the proposed approach we have constructed a model crystal with a cubic lattice that consists of three different types of atoms. This model crystal contained 8000 atoms. Masses of atoms correspond by convention to  $Si^{28}$ ,  $Ba^{145}$  and  $U^{238}$ . Further we denote these atoms as  $m_1$ ,  $m_2$  and  $m_3$ , respectively. In Figure 1 a fragment of the model crystal is shown. The occupation of the lattice sites was chosen in such a way that each U or Ba atom has light Si atoms as nearest neighbors and each U (Ba) atom has Ba (U) atoms as second nearest neighbors. In our calculations, Lenard-Jones potential [21] was used with parameters taken from [22-24]. These parameters were slightly varied to stabilize the lattice of the model crystal. The potentials that describe the interaction of atoms of different types were taken as an average of constituent's interatomic potentials. Further simulations were devoted to the study of the influence of the ratio of atomic masses on the radiation effect under the bombardment of model crystal by Nitrogen (N) ions with energy in different energy intervals.

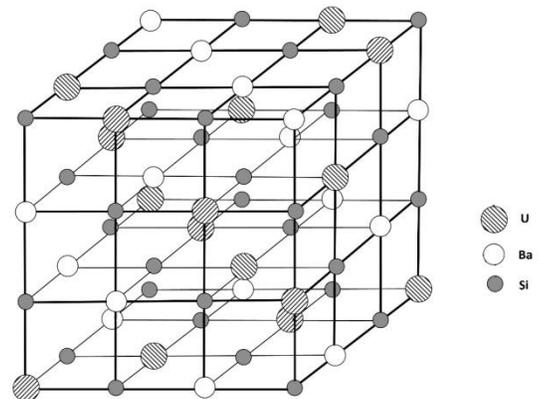


Figure1. Fragment of the model crystal.

In computer experiment the bombardment was performed in two energy (E) intervals: (I) 34 eV – 93 eV and (II) 93 eV – 142 eV determined according to expression (1).

It is clear that the ion bombardment in the energy interval (I) leads to displacement of only Si atoms into the

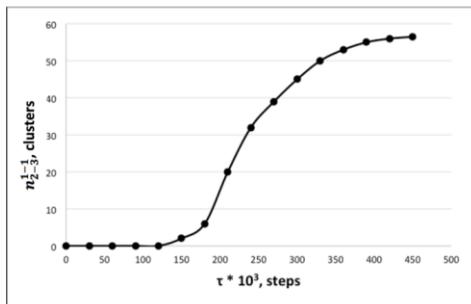
interstitial positions whereas a bombardment in the interval (II) leads to such displacement of both Si and Ba atoms.

#### 4. Ion bombardment of triatomic crystal. Results and discussion

We started with the consideration of the action of particles in the first energy interval on radiation-induced structural changes in the target crystal. As a result of simulations it was found that the formation of clusters of heavier atoms (U and Ba) occurs. According to our definition, in *heavy clusters* heavy atoms occupy the nearest neighbor positions in the lattice. Such clusters are formed due to radiation-induced formation of vacancies in the sublattice of light atoms (Si). These vacancies stimulate the displacements of heavy atoms. As a result U and Ba move closer to each other forming heavy clusters. Radiation stimulated diffusion of heavy atoms under the bombardment of particles with energy less than the threshold energy of the atomic displacement also promotes the formation of such clusters.

The kinetics of the accumulation of clusters is shown in Figure 2. It may be seen that for the formation of heavy clusters some threshold dose exists. A consideration of atomic configurations in the model crystal showed that at higher doses the formation of three-atomic clusters starts by adding to diatomic clusters ( $m_2$ - $m_3$ ) one additional atom with the mass  $m_2$ .

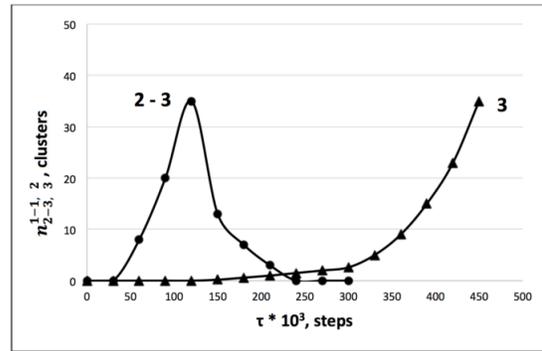
This explains the saturation of the formation of diatomic ( $m_2$ - $m_3$ ) clusters demonstrated by the kinetic curve. With the further increase of the dose the accumulation of clusters containing  $m_2$  and  $m_3$  atoms stops and clusters containing only  $m_3$  atoms are formed.



**Figure 2.** Kinetics of accumulation of  $m_2$ - $m_3$  clusters (the bombardment by ions in the first energy interval).

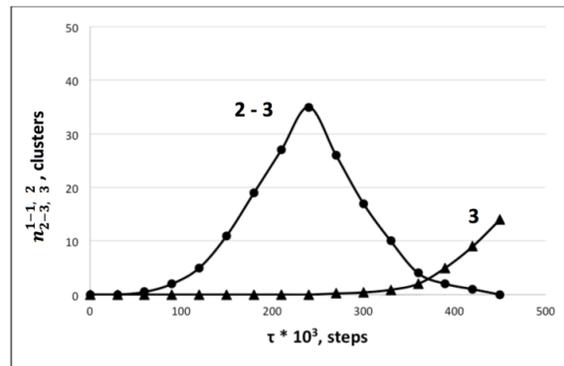
Here  $n_{2-3}^{1-1}$  is the number of diatomic clusters formed by one atom with mass  $m_2$  and one atom with mass  $m_3$ .

As a result of a bombardment of the model crystal by ions with the energy in the second energy interval we observed a formation of similar diatomic  $m_2$ - $m_3$  clusters which are further destroyed with accumulation of dose. Simultaneously with the destruction of  $m_2$ - $m_3$  clusters the diatomic clusters containing only atoms with the mass  $m_3$  are formed as displayed in Figure 3.



**Figure 3.** Kinetics of diatomic clusters formed by atoms with the masses  $m_2$  and  $m_3$  and by atoms with the mass  $m_3$  (the bombardment by ions in the second energy interval, 300 hits/step). Here  $n_{2-3}^{1-1}$  is the same as in Figure 2, and  $n_3^2$  is the number of diatomic clusters formed by atoms with the mass  $m_3$ . Circles and triangles correspond to  $n_{2-3}^{1-1}$  and  $n_3^2$ , respectively.

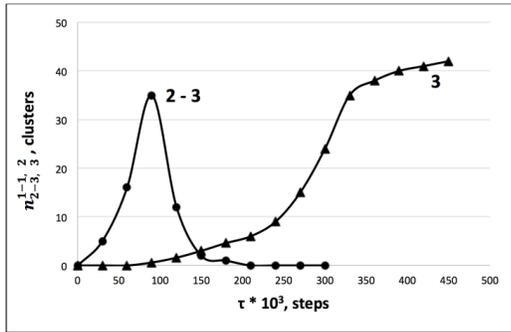
We obtained that the kinetics presented in Figure 3 changes when the intensity of the ion beam is changed. As already mentioned, in our model the intensity of the ion beam is proportional to the number of atoms that get a hit during one calculation step. Figure 4 illustrates the results of formation of similar  $m_2$ - $m_3$  and  $m_3$  diatomic clusters in the case of lower intensity of the ion beam than in the case shown in Figure 3. It can be seen that the formation of  $m_2$ - $m_3$  clusters starts later and their life time is longer. Respectively, the diatomic clusters of the atoms with the mass  $m_3$  form also later.



**Figure 4.** Kinetics of diatomic clusters  $m_2$ - $m_3$  and  $m_3$  (the bombardment by ions in the second energy interval, 200 hits/step) Circles and triangles correspond to  $n_{2-3}^{1-1}$  and  $n_3^2$ , respectively.

Figure 5 illustrates that opposite conclusions are derived for the intensity of the ion beam that is in half larger than in the case shown in Fig. 3. As follows from Figure 5 the formation of  $m_2$ - $m_3$  clusters begins almost immediately after the irradiation starts and they are stable for a shorter period of time. Diatomic clusters, consisting of atoms with the

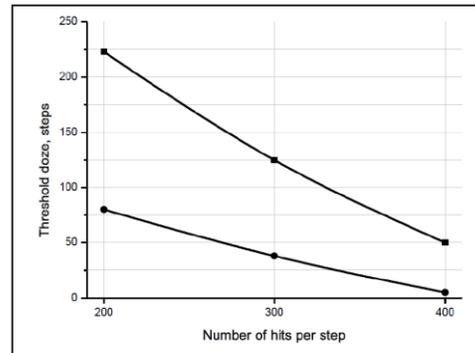
large mass  $m_3$ , start to form much earlier than in the case shown in Fig. 3.



**Figure 5.** Kinetics of clusters  $m_2$ - $m_3$  and  $m_3$  (the bombardment by ions in the second energy interval, 400 hits/step) Circles and triangles correspond to  $n_{2-3}^{1-1}$  and  $n_3^2$ , respectively.

Figure 5 shows that at larger dose rate the kinetics of accumulation of diatomic clusters of  $m_3$  atoms tends to saturate. At these doses it was observed that three-atomic and four-atomic clusters of  $m_3$  atoms start to form. It is obvious that they are formed not only by bringing together three or four massive atoms but also by attachment of additional atoms to already existing diatomic clusters of heavy atoms. So, the saturation stage of the kinetics of  $m_3$  clusters may be reasonably explained by transition of diatomic clusters of atoms with the mass  $m_3$  into polyatomic clusters of these atoms. It was mentioned that in the second energy interval (93 eV – 142 eV) the atoms with masses  $m_1$  and  $m_2$  are displaced from the lattice sites. At some dose these clusters are destroyed and two-atomic clusters, consisting only of atoms with the mass  $m_3$ , are formed. This is caused due to the ability of bombarding particles (in this energy interval) to displace  $m_2$  atoms. Further we observed the increase of the number of atoms of  $m_3$  type in the heavy clusters.

Figs. 6-8 demonstrate the dependence of the characteristic parameters of the formation and destruction of heavy clusters on the intensity of the ion beam. Fig. 6 depicts the dependence of the threshold dose for  $m_2$ - $m_3$  clusters formation under the ion irradiation in the first and in the second energy intervals on the dose rate of irradiation.

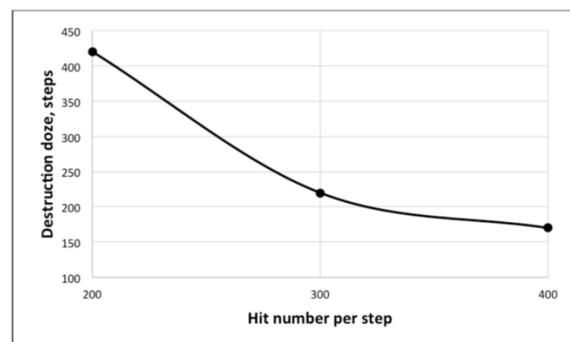


**Figure 6.** Dependence of the threshold dose for the formation of diatomic  $m_2$ - $m_3$  clusters on the dose rate under irradiation by ions in the first (squares) and second (circles) energy intervals.

The almost linear dependences of threshold doses for heavy clusters formation on the dose rate means that the accumulation of vacancies leads to significant decrease in the bonding of heavy atoms in the lattice. In its turn, this changes the equilibrium positions of heavy atoms in the lattice and leads to the formation of heavy clusters.

Fig 7 shows how the dose at which the  $m_2$ - $m_3$  clusters break up depends on the intensity of the ion beam. Fig. 8 illustrates the dependence of the threshold dose for the formation of diatomic clusters of  $m_3$  atoms under irradiation by ions in the second energy interval on the dose rate. The non-linear dependences on the dose rate of the threshold doses for  $m_2$ - $m_3$  clusters destruction and  $m_3$  clusters formation in this case are explained by the necessity of the preliminary breaking of  $m_2$ - $m_3$  clusters by incident particles.

Analysis of results of triatomic model crystal bombardment by incident particles from the second energy interval allows suggesting the features of the formation of clusters at higher energies. In this case, as a result of elastic collisions, all three types of atoms can be displaced. This facilitates the formation of different heavy clusters. Although simultaneously heavy clusters will collapse, the heaviest clusters will collapse to less extent. Thus, the radiation effect will lead to the predominance of the heaviest clusters.



**Fig.7.** Dependence of the destruction dose on the dose rate for  $m_2$ - $m_3$  clusters under irradiation by ions in the second energy interval.

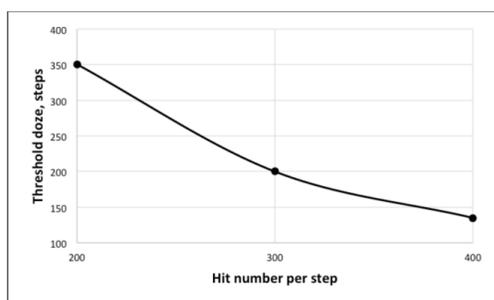


Fig. 8. Dependence of the formation dose on the dose rate for  $m_3$  clusters under irradiation by ions in the second energy interval.

## 5. Summary

We report the MD approach that allows clarification of the contribution of incident particles in the selected interval of energy spectrum to the final structural damage of the target crystal. Its application for the case of triatomic model crystal is demonstrated. A formation of clusters caused only by atomic elastic collisions is considered. The results obtained for model crystal are of a general nature, and the developed approach can be applied for study of radiation effects in different materials including recently reported U-Al-Si phases [25].

## Acknowledgements

This study was supported by the joint IAEC-UPBC Pazy foundation.

## References

[1] T. Schlick, *Molecular Modeling and Simulation*, Springer -Verlag, Berlin – Heidelberg - New York, 2002.  
 [2] D. Frenkel, B. Smit, *Understanding Molecular Simulation* (Second ed.), Academic Press, San Diego, 2002.  
 [3] F. Ercolessi, *A molecular dynamics primer*, Spring College in Computational Physics, ICTP, Trieste, 1997.  
 [4] D. C. Rapaport, *The art of molecular dynamics simulation*, Cambridge University Press, UK, 2004.  
 [5] Akira Satoh, *Introduction to practice of molecular simulation*, Akita Prefectural University, Japan, 2010.  
 [6] J. Edward, Jr. Smiley, *Molecular Dynamics Simulations of Atomic and Cluster Bombarded Surfaces*, The Pennsylvania State University, School of chemistry, USA, 2006.  
 [7] R. S. Nelson, *The Observation of Atomic Collisions in Crystalline Solids*, Series “Defects in crystalline solids” (Eds. S. Amelinskx, R. Gevers, J. Nihoul), North-Holland Publishing Company, Amsterdam, 2013.  
 [8] P. Derlet, M. D. Nguyen-Manh, and S. L. Dudarev, Multiscale modeling of crowdion and vacancy defects in body-centered-cubic transition metals, *Phys. Rev.*, B 76, pp. 054107 -12, 2007.  
 [9] N. Mykytenko, D. Fink, A. Kiv, *Journal Comp. Sci.*, 6, pp. 34-39, 2015.

[10] D. Fink, A. Petrov, W. R. Fahrner, et al., Ion Track-Based Nanoelectronics, *Int. J. Nanoscience* 4, pp. 965 -73, 2005.  
 [11] D. Fink, L. T. Chadderton, K. Hoppe, et al., Study of ferrofluids in confined geometry, *Nucl. Instr. Meth. Physical Res.*, B 26, pp. 727–730, 2007.  
 [12] K. Hoppe, D. Fink, W. R. Fahrner, Metallized Nuclear Tracks in Quasi Metal Oxide Semiconductor Structures for Electronic Devices, *J. Electrochem. Soc.*, 155, pp. 7 – 11, 2008.  
 [13] D. Fink, I. Klinkovich, O. Bukelman, R. S. Marks, , A. Kiv, D. Fuks, W. R. Fahrner, L. Alfonta, Glucose determination using a re-usable enzyme-modified ion track membrane sensor *Biosensors and Bioelectronics*, 24, pp. 2702-6, 2009.  
 [14] J. Honerkamp, *Statistical Physics: An Advanced Approach with Applications*, 3rd ed., Springer, Germany, 2012.  
 [15] J. Harris, , Some notes on multiplicative congruential random number generators with Mersenne prime modulus  $2^{61} - 1$ , *Journal of the South Carolina Academy of Science*, 1 pp. 28-32, 2003  
 [16] P. Viot, *Numerical Simulation in Statistical Physics Lecture in Master 2: “Physics of complex systems” and “Modeling, Statistics and Algorithms for out-of-equilibrium systems”*, Laboratoire de Physique Theorique de la Matiere Condensee, Paris, 2014.  
 [17] J. M. Haile, *Molecular Dynamics Simulation: Elementary Methods*, Wiley, USA, 1997.  
 [18] M. Griebel, S. Knapek, G. Zumbusch, *Numerical Simulation in Molecular Dynamics*, Springer, Bonn, 2007.  
 [19] S. Parviainen, F. Djurabekova, H. Timko, K. Nordlund, Electronic processes in molecular dynamics simulations of nanoscale metal tips under electric fields, *Comput. Mat. Sci.*, 50, pp. 2075–2079, 2011.  
 [20] E. Hairer, Ch. Lubich, G. Wanner, Geometric numerical integration illustrated by the Stormer-Verlet method, *Acta Numerica*, 1, p. 399–450, 2003.  
 [21] S. Zhen, G. J Davies, Calculation of the Lennard-Jones n-m potential energy parameters for metals, *Phys. Stat. sol.* (a), 78, pp. 595- 605, 1983.  
 [22] S. M. Ghiaasiaan, *Convective Heat and Mass Transfer, Appendix K: Lennard Jones Potential Model Constants for Selected Molecules*, Cambridge University Press, New York, 2011.  
 [23] M. Damyanova, L. Zarkova, U. Hohm, Effective Intermolecular Interaction Potentials of Gaseous Fluorine, Chlorine, Bromine, and Iodine, *Int. Thermophys*, 30, pp. 1165-1178, 2009.  
 [24] Edith Beerdsen, David Dubbeldam, Berend Smit, Simulating the Effect of Nonframework Cations on the Adsorption of Alkanes in MFI-type Zeolites, *J. Phys. Chem. B*, 107, pp. 12088 – 12096, 2003.  
 [25] Rafailov G., Dahan I., Meshi L., New ordered phase in the quasi-binary  $UAl_3 - USi_3$  system, *Acta Cryst. B*, 70, pp. 3580-585, 2014.

## Biographies

**Arnold Kiv** received the D. Sc. (Dr. Hab.) degree in solid state physics from Tartu Institute of Physics, Tartu, Estonia, in 1978. From 1964 to 1982, he was a Senior Researcher and a Head of the Laboratory of Radiation Effects, Institute of Nuclear Physics, Academy of Sciences, Tashkent. From 1983 to 1998, he was a Head of the Department of Theoretical Physics, South-Ukrainian National Pedagogical University, Odessa, Ukraine. In 1997, he was an Invited Professor, Western Ontario University, Canada. From 1999 to the present, he is a Professor-Researcher in the Department of Materials Engineering, Ben-Gurion University of the Negev, Israel. In 1996 and 2011 he was co-Director of NATO Advanced research Workshops and an Editor of two NATO Series books. He has about 300 publications, three monographs and three Invention Certificates in the field of computer simulation of radiation effects in solid state electronics. His research interests include mechanisms of formation of radiation defects in solids, interaction of fast particles with materials, including computer simulation and experimental studies. **E-mail: kiv@bgu.ac.il**

**Natalia Mykytenko** received her B. Sci. Degree in 2008 from Odessa I. I. Mechnikov National University in the field of computing systems and networks and M. Sci. Degree in the same field in 2010. At present she is a post graduate student at the Department of Physical and Mathematical Modeling of South-Ukrainian national pedagogical university and Researcher-Programmer in Hewlett –Packard Company. Her scientific interests are in the direction of the computer modeling of radiation defects and diffusion processes in solids. She published over 10 peer-reviewed papers on these topics. **E-mail: mykytenkon@gmail.com**

**David Fuks** received the Ph.D. degree in solid state physics from Tomsk State University, Tomsk, Russia, and the D.Sc. degree (Dr. Hab.) from Moscow State University, Moscow, Russia, in 1975 and 1984, respectively. In 1999, he was Invited Professor in the Institute of Physics, Federal University of Bahia, Bahia, Brazil, where he started his research on molecular dynamics simulations in solids. He was also Invited Professor in the Department of Metallurgy and Materials Science, Catholic University of Leuven, Leuven, Belgium, continuing this research. In 2000, he was Invited Professor in the Physics Department, Osnabrück University, and Osnabrück, Germany where his research was devoted to the study of electronic properties of ferroelectrics. He has about 250 publications His main research interests concern quantum-mechanical theory of solids; molecular dynamics simulations, and computer modeling of processes of charge storage in dielectric media. **E-mail: fuks@bgu.ac.il**

**Itzchak (Isaac) Dahan** received his Ph.D. in materials engineering from Ben Gurion University (BGU) of the Negev, Beer Sheva, Israel in 2005, specializing on metallurgy of thin films. Dr. Dahan spent two years sabbatical in Colorado School of Mines (CSM), Golden CO, USA, during which he specialized in the study of structural coatings, mainly Uranium oxides ( $\text{UO}_2$  and  $\text{U}_3\text{O}_8$ ). He joined metallurgical group in NRCN, Israel at 1998 as a researcher, and then he became the head of the group at 2003. His main research interests are metallurgy of Uranium alloys, investigation of phase transformation in these alloys, influence of inclusions and precipitates on the mechanical and physical properties, and, correlation between them. Dr. Dahan has published over 20 peer-reviewed papers on topics listed above. **E-mail: idahan60@gmail.com**

**Louisa Meshi** received her Ph.D. in materials engineering from Ben Gurion University (BGU) of the Negev, Beer Sheva, Israel in 2006, specializing on structure determination of intermetallic compounds using a combination of powder X-ray diffraction and electron crystallography methods. During her Ph.D. studies she received two prestigious awards: The Lev Margulis prize of the Israeli Society for Microscopy and the Wolf prize for excellent doctoral research. Following postdoctoral research in Bristol University, UK, during which she specialized in the study of structural defects using electron microscopy, Dr. Meshi has joined in 2009 the BGU Materials Engineering Department. Her research interests include determination of atomic structure of novel materials, study of structural defects and phase transformations. Dr. Meshi has published over 45 peer-reviewed papers on these topics. In 2012, Dr. Meshi received the Krill Award of the Wolf foundation for excellence in scientific research. **E-mail: louisa.meshi@nist.gov**